

## On the Arsenides and Antimonides of Tantalum

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The phase relationships in the systems tantalum-arsenic and tantalum-antimony have been studied by X-ray method. Density determinations and magnetic susceptibility measurements have been carried out. Five intermediate phases have been identified:

1. TaAs, with NbAs-type structure,  $a = 3.4348 \text{ \AA}$ ,  
 $c = 11.641 \text{ \AA}$ , density =  $12.25 \text{ g cm}^{-3}$ .
2. TaAs<sub>2</sub>, with NbAs<sub>2</sub>-type structure,  $a = 9.3385 \text{ \AA}$ ,  
 $b = 3.3851 \text{ \AA}$ ,  $c = 7.7568 \text{ \AA}$ ,  $\beta = 119.70^\circ$ , density =  $10.26 \text{ g cm}^{-3}$ .
3. Ta<sub>3</sub>Sb, with  $\beta$ W-type structure,  $a = 5.2646 \text{ \AA}$ .
4. Ta<sub>3</sub>Sb<sub>4</sub>, with Ti<sub>3</sub>Te<sub>4</sub>-type structure,  $a = 10.248 \text{ \AA}$ ,  
 $c = 3.5460 \text{ \AA}$ , density =  $12.22 \text{ g cm}^{-3}$ .
5. TaSb<sub>2</sub>, with NbAs<sub>2</sub>-type structure,  $a = 10.2218 \text{ \AA}$ ,  
 $b = 3.6447 \text{ \AA}$ ,  $c = 8.2915 \text{ \AA}$ ,  $\beta = 120.39^\circ$ , density =  $10.53 \text{ g cm}^{-3}$ .

The TaAs, TaAs<sub>2</sub>, and TaSb<sub>2</sub> phases have diamagnetic susceptibilities, whereas the Ta<sub>3</sub>Sb<sub>4</sub> phase shows weak temperature independent paramagnetism.

In the course of continued studies on transition metal chalcogenides and pnictides at this Institute an investigation of structural and magnetic properties of tantalum arsenides and antimonides has been carried out. A study of these systems has been prompted by interesting structural findings in the corresponding niobium-arsenic and niobium-antimony systems.<sup>1-4</sup>

The present communication forms a supplement to recent publications on the tantalum arsenides by Boller and Parthé<sup>5</sup> and Saini *et al.*<sup>6</sup>, where phases with composition TaAs and TaAs<sub>2</sub> have been described. The older literature<sup>7</sup> claims the existence of a phase with composition TaAs<sub>1.4</sub>. Ta<sub>3</sub>Sb is the only phase previously reported<sup>8</sup> in the tantalum-antimony system.

### EXPERIMENTAL

*Materials.* The tantalum metal used in this study was "Spectrographically standardized tantalum" from Johnson, Matthey & Co., Ltd.; with reported impurities (in ppm): Nb (<200), Fe (300), Si (100), Al (20), Cr (20), Sn (20), Mn (5), and Mg (1). The "Spectrographically standardized" arsenic and antimony from Johnson, Matthey & Co., Ltd. contained (in ppm): Na (3), Si (2), Cu (<1), Mg (<1), and Ag (<1) for the arsenic and Cu (<1), Mg (<1), Si (<1), and Ag (<1) for the antimony.

*Preparation.* Samples were prepared by heating accurately weighed quantities of tantalum and arsenic or antimony, respectively, in evacuated and sealed silica tubes. Samples with composition 33.33, 50.00, 66.67, and 75.00 atomic % As and samples with 20.00, 25.00, 33.33, 44.44, 50.00, 60.00, 66.67, and 75.00 atomic % Sb were heated at 850°C for 10 days and quenched in ice water.

Samples of TaAs, Ta<sub>3</sub>Sb, and Ta<sub>5</sub>Sb<sub>4</sub> were also prepared by thermal decomposition of TaAs<sub>2</sub> and TaSb<sub>2</sub>, respectively.

*X-Ray diffraction.* All samples were crushed and X-ray photographs were taken in a Guinier focusing camera of 80 mm diameter using strictly monochromatized CuK $\alpha_1$ -radiation. For the calculation of lattice constants potassium chloride (Analar, The British Drug Houses Ltd.,  $a = 6.2919 \text{ \AA}$ ) was added to the specimen as an internal standard. Lattice constants are expressed in Ångström units on the basis of  $\lambda(\text{CuK}\alpha_1) = 1.54050 \text{ \AA}$ .

*Density measurements.* The density measurements were carried out pycnometrically at 25°C with kerosene as displacement liquid. To remove gases adsorbed by the sample the pycnometer was filled under vacuum with kerosene. The samples weighed approximately 2 g.

*Magnetic measurements.* The magnetic susceptibilities were measured by the Gouy method at three different maximum field strengths ( $H_{\text{max}} = 4015, 4700, \text{ and } 5110 \text{ O}$ ). The samples were enclosed in evacuated and sealed Pyrex tubes of 3 mm internal diameter to a height of about 85 mm.

## RESULTS

In the tantalum-arsenic system only the two intermediate phases TaAs and TaAs<sub>2</sub>, previously described,<sup>5,6</sup> were found. In the tantalum-antimony system the existence of two new, intermediate phases was ascertained, *i.e.* the Ta<sub>5</sub>Sb<sub>4</sub> phase and the TaSb<sub>2</sub> phase. Furthermore, the existence of the Ta<sub>3</sub>Sb phase (known from the earlier study of Nevitt<sup>8</sup>) was confirmed.

These phases are all completely analogous to the phases in the niobium-arsenic and niobium-antimony systems, as described by Furusetth and Kjekshus.<sup>1-4</sup>

*The Ta<sub>3</sub>Sb phase.* Guinier photographs of the samples with < 44.44 atomic % Sb showed the presence of a phase with cubic symmetry. The indexed reflections are no doubt due to the Ta<sub>3</sub>Sb phase, described by Nevitt.<sup>8</sup>

The purest samples of the Ta<sub>3</sub>Sb phase were obtained by thermal decomposition of the TaSb<sub>2</sub> phase at 1150°C. After degradation the residuum was found to contain Ta<sub>3</sub>Sb as the only phase from the tantalum-antimony system. However, a parallel, interfering reaction with silica gave the samples an unavoidable contamination. (A similar observation is made in a study of tantalum chalcogenides with high tantalum content.<sup>10</sup>)

The lattice constant of the Ta<sub>3</sub>Sb phase (the purest sample judging from the Guinier photograph (Table 1)) is  $a = 5.2646 \text{ \AA}$  in close agreement with the value  $a = 5.2595 \pm 0.0010 \text{ \AA}$  reported by Nevitt.<sup>8</sup> According to the approximately constant  $a$ -axis for samples of different composition, the homogeneity range of the Ta<sub>3</sub>Sb phase is not noticeable.

The observed density is approximately 10 % lower than the calculated density, 15.12 g cm<sup>-3</sup>, from the X-ray measurements assuming 2 Ta<sub>3</sub>Sb-groups per unit cell. This discrepancy is readily explained by the contamination of the sample and gives no reason to reject the stoichiometric formula Ta<sub>3</sub>Sb.

According to Nevitt<sup>8</sup> Ta<sub>3</sub>Sb should be listed among substances having  $\beta W$ -type structure. A comparison of observed and calculated intensities of the

Table 1. Guinier photograph data of Ta<sub>3</sub>Sb taken with strictly monochromatized CuK $\alpha_1$  radiation.

$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$	$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$
	obs	calc			obs	calc	
vw	4 286	4 281	110	st	27 839	27 828	320
m	8 563	8 562	200	m	34 255	34 250	400
vst	10 699	10 703	210	m	42 821	42 812	420
st	12 843	12 844	211	st	44 969	44 953	421
m	25 679	25 687	222	st	47 144	47 093	332

reflections on the Guinier photographs confirmed the proposed structure. The shortest interatomic distances in Ta<sub>3</sub>Sb are (in Å):

$$\begin{aligned} \text{Ta} - 10 \text{ Ta: } & 2.632 (2); 3.224(8) \\ & - 4 \text{ Sb: } 2.943 \\ \text{Sb} - 12 \text{ Ta: } & 2.943 \\ & - 8 \text{ Sb: } 4.559 \end{aligned}$$

The shortest Ta-Ta distance is some 8 % shorter than in metallic tantalum (2.863 Å<sup>11</sup>). Similar short metal-metal distances observed in all phases with  $\beta$ W-type structure, are one of the most interesting features of this class of phases (see the discussion by Geller<sup>12</sup> and Pauling;<sup>13</sup> see also Nevitt<sup>8</sup>).

The Ta<sub>5</sub>Sb<sub>4</sub> phase. The disappearing phase method applied to the Guinier photographs indicated the existence of a phase with composition 44.44

Table 2. Guinier photograph data of Ta<sub>5</sub>Sb<sub>4</sub> taken with strictly monochromatized CuK $\alpha_1$  radiation.

$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$	$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$
	obs	calc			obs	calc	
st	1 131	1 130	110	w	20 315	20 335	006
w	2 259	2 259	200	w	21 100	{ 21 099	521
w	4 524	4 519	220			{ 21 132	202
vw	5 643	5 649	310	m	22 585	22 595	620
st	7 542	7 542	211	vw	23 397	23 392	222
vst	9 789	9 802	301	w	25 637	25 618	611
st	10 168	10 168	330	st	27 929	{ 27 878	541
st	11 302	11 297	420			{ 27 911	402
m	12 056	12 061	321	m	28 238	28 244	{ 710
vst	14 313	14 321	411				{ 550
m	14 685	14 687	510	m	29 055	29 041	332
vw	18 065	18 076	440	st	30 183	30 170	422
		{ 18 840	{ 431	w	33 573	33 560	512
m	18 867	{ 18 873	{ 501	st	34 670	34 656	721
w	19 193	19 206	530	vw	36 134	36 152	800
vw	20 010	20 003	112	vw	36 979	36 949	442

Table 3. Interatomic distances in  $Ta_5Sb_4$  (Å).

$Ta_I$	— 2 $Ta_I$	: 3.546 (2)
	— 8 $Ta_{II}$	: 2.91 (8)
	— 4 Sb	: 2.93 (4)
$Ta_{II}$	— 2 $Ta_I$	: 2.91 (2)
	— 6 $Ta_{II}$	: 3.26 (2); 3.43 (2); 3.546 (2)
	— 5 Sb	: 2.72 (2); 2.76 (3)
Sb	— 1 $Ta_I$	: 2.93
	— 6 $Ta_{II}$	: 2.72 (2); 2.76 (4)
	— 10 Sb	: 3.546 (2); 3.75 (4); 4.15 (2); 4.32 (2)

atomic % Sb. Thermal decomposition of the  $TaSb_2$  phase at 1000°C confirmed this composition and the phase was accordingly designated  $Ta_5Sb_4$ . Density measurements and structure determination proved the composition.

The Guinier photographs (Table 2) were indexed on tetragonal axes:

$$a = 10.248 \text{ \AA}, c = 3.5460 \text{ \AA}, c/a = 0.3460$$

The lattice constants of the stoichiometric  $Ta_5Sb_4$  sample vary only slightly from the lattice constants of  $Ta_5Sb_4$  in samples where the  $Ta_5Sb_4$  phase is in equilibrium with the  $Ta_3Sb$  and  $TaSb_2$  phases. This indicates that the homogeneity range of the  $Ta_5Sb_4$  phase is rather narrow.

According to the pycnometric density  $12.25 \text{ g cm}^{-3}$  the unit cell contains 2 ( $Z_c = 1.97$ )  $Ta_5Sb_4$ -groups. The calculated density from the X-ray data is  $12.41 \text{ g cm}^{-3}$ .

The tetragonal structure of a phase  $Ti_5Te_4$  has been described by Grønvold *et al.*<sup>14</sup> As it has been shown<sup>1,2</sup> that the compounds  $Nb_5Sb_4$  and  $Ti_5Te_4$  are isostructural, and as the Guinier photograph of  $Ta_5Sb_4$  resembles that of  $Nb_5Sb_4$  both with regard to line pattern and intensities, the possibility of  $Ta_5Sb_4$  also being isostructural with  $Ti_5Te_4$  was investigated.

In terms of the space group  $I4/m$  the atomic arrangement in  $Ti_5Te_4$  is as follows (the parameters have been rounded off):

$$\begin{aligned} 2 \text{ Ti}_I & \text{ in } (a) \ 0,0,0; \frac{1}{2}, \frac{1}{2}, \frac{1}{2} \\ 8 \text{ Ti}_{II} & \text{ in } (h) \ \pm(x,y,0; \frac{1}{2}+x, \frac{1}{2}+y, \frac{1}{2}; \bar{y}, x, 0; \frac{1}{2}-y, \frac{1}{2}+x, \frac{1}{2}) \\ & \text{ with } x_1 = 0.31 \text{ and } y_1 = 0.38 \\ 8 \text{ Te} & \text{ in } (h) \ \text{with } x_2 = 0.06 \text{ and } y_2 = 0.28 \end{aligned}$$

Table 4. Guinier photograph data of TaAs taken with strictly monochromatized  $CuK\alpha_1$ -radiation.

$I_{obs}$	$\sin^2\theta \times 10^6$		$hkl$	$I_{obs}$	$\sin^2\theta \times 10^6$		$hkl$
	obs	calc			obs	calc	
m	5 462	5 466	101	w	25 582	25 580	211
m	6 995	7 004	004	w	25 819	25 817	116
m	8 965	8 969	103	w	26 487	26 480	107
vst	11 806	11 808	112	m	27 118	27 118	204
w	15 968	15 973	105	vw	28 027	28 018	008
m	20 107	20 114	200	w	29 083	29 083	213

Table 5. Lattice constants of TaAs.

$a$ (Å)	$c$ (Å)	$c/a$	Reference
3.4348	11.641	3.3891	Present
3.43 <sub>7</sub>	11.65 <sub>6</sub>	3.39 <sub>1</sub>	Boller and Parthé <sup>5</sup>
3.436 ± 0.002	11.644 ± 0.005	3.389	Saini <i>et al.</i> <sup>6</sup>

These parameters were used in the calculation of intensities of the Guinier photographs. A reasonable agreement between observed and calculated intensities was obtained, indicating that the proposed structure is correct. Further refinement of the parameters was not attempted.

The shortest interatomic distances, based on the above parameters, are listed in Table 3. A discussion of this structure type is given by Grønvold *et al.*<sup>14</sup>

Seven phases are known with the  $Ti_5Te_4$ -type structure, *i.e.*  $Ti_5Te_4$ ,<sup>14</sup>  $V_5S_4$ ,<sup>15</sup>  $V_5Se_4$ ,<sup>16</sup>  $Nb_5Se_4$ ,<sup>17</sup>  $Nb_5Te_4$ ,<sup>17</sup>  $Nb_5Sb_4$ ,<sup>1,2</sup> and  $Ta_5Sb_4$  while  $V_5Te_4$ <sup>18</sup> has a monoclinic structure of similar dimensions. The existence of  $Nb_5Sb_4$  and  $Ta_5Sb_4$  emphasizes the metallic character of these phases as already pointed out by Furuseth and Kjekshus.<sup>2</sup>

*The TaAs phase.* The purest samples of the TaAs phase were made by thermal decomposition of the  $TaAs_2$  phase. Complete degradation of  $TaAs_2$  at 900°C gave TaAs as residual crystalline phase.

Guinier photographs (Table 4) of the TaAs phase were indexed on the basis of a tetragonal unit cell. The lattice dimensions of TaAs being essentially constant for samples with lower and higher arsenic content than stoichiometric TaAs, show that the homogeneity range must be rather narrow.

For comparison with the values previously determined by Boller and Parthé<sup>5</sup> and Saini *et al.*<sup>6</sup> the lattice constants have been listed in Table 5. The three sets of values are in good agreement.

The pycnometric density of TaAs prepared by thermal decomposition of  $TaAs_2$ , 12.25 g cm<sup>-3</sup>, is in close accordance with the calculated density, 12.37 g cm<sup>-3</sup>, from the X-ray data (assuming 4 ( $Z_c = 3.96$ ) TaAs-groups per unit cell).

The crystal structure of TaAs has been determined by Boller and Parthé.<sup>5</sup> (A tentative confirmation is also given by Saini *et al.*<sup>6</sup>) TaAs is isostructural with NbAs, a structure independently determined by Boller and Parthé<sup>5</sup> and Furuseth and Kjekshus.<sup>3</sup> In terms of space group  $I4_1md$  4 Ta and 4 As are in ( $a$ ):

$$0, 0, z; 0, \frac{1}{2}, \frac{1}{4} + z; \frac{1}{2}, 0, \frac{3}{4} + z; \frac{1}{2}, \frac{1}{2}, \frac{1}{2} + z$$

with  $z_{Ta} = 0$  and  $z_{As} \approx 5/12$

This structure was also verified in the present study by comparing observed and calculated intensities of the Guinier photographs, though minor deviations from  $z_{As} = 5/12$  were not tested.

Table 6. Guinier photograph data of TaAs<sub>3</sub> taken with strictly monochromatized CuK $\alpha_1$ -radiation.

$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$	$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$
	obs	calc			obs	calc	
m	1 300	1 307	001	w	27 702	27 717	510
st	2 758	2 762	20 $\bar{1}$	w	27 768	27 774	221
w	3 598	3 606	200	w	28 238	28 255	402
m	4 525	4 532	20 $\bar{2}$	w	29 619	29 624	22 $\bar{3}$
vw	5 220	5 227	002	m	29 811	29 828	31 $\bar{5}$
m	6 072	6 079	110	m	31 282	31 282	114
m	6 305	6 310	11 $\bar{1}$	w	31 764	31 758	42 $\bar{2}$
w	7 054	7 064	201	w	32 147	32 140	42 $\bar{1}$
vst	8 458	8 461	111	m	32 473	{32 457	600
m	8 905	8 914	20 $\bar{3}$	m		{32 470	023
st	9 143	9 155	11 $\bar{2}$	w	32 854	32 862	60 $\bar{5}$
w	11 041	11 049	40 $\bar{2}$	w	33 108	33 117	204
w	11 369	11 372	31 $\bar{1}$	vw	33 380	33 370	11 $\bar{5}$
m	11 422	11 430	40 $\bar{1}$	vw	33 487	33 499	51 $\bar{5}$
st	11 751	11 761	003	m	33 986	33 990	42 $\bar{3}$
st	12 059	12 066	31 $\bar{2}$	m	34 396	34 401	511
vst	13 272	{13 281	40 $\bar{3}$	w	34 740	34 731	313
st	13 444	{13 292	310	m	35 148	35 135	420
m	14 420	13 457	112	vw	35 653	35 657	40 $\bar{6}$
m	14 605	14 425	400	w	36 638	36 620	224
st	15 361	14 613	11 $\bar{3}$	w	37 747	37 743	20 $\bar{6}$
m	15 897	15 373	31 $\bar{3}$	st	38 536	38 531	71 $\bar{3}$
vw	17 811	15 911	204	vw	38 850	38 836	424
vw	18 125	17 825	311	vw	39 097	39 091	403
w	20 028	18 126	404	vw	40 156	{40 150	714
st	20 696	20 034	401			{40 216	601
vw	21 043	20 696	020	w	40 764	{40 743	421
w	21 282	21 066	113	w	40 996	{40 784	60 $\bar{6}$
m	21 802	21 294	314	w	40 996	40 976	31 $\bar{6}$
vw	22 004	21 819	203	st	42 532	{42 496	51 $\bar{6}$
m	22 171	22 017	021	vw	43 136	{42 529	223
vw	22 666	22 189	51 $\bar{2}$	vw	43 136	43 133	71 $\bar{1}$
w	23 334	22 685	114	vw	43 690	{43 651	803
w	23 471	23 346	51 $\bar{3}$			{43 698	512
vw	23 634	23 472	22 $\bar{1}$	m	44 149	{44 124	115
vw	24 312	23 646	51 $\bar{1}$			{44 195	804
vw	24 766	24 316	220	vw	44 401	44 383	71 $\bar{5}$
vw	24 962	24 778	60 $\bar{2}$	w	45 503	{45 488	62 $\bar{2}$
w	25 238	24 971	312	w	45 503	{45 569	62 $\bar{3}$
w	25 497	25 241	22 $\bar{2}$	w	45 742	45 721	802
vw	25 935	25 520	20 $\bar{5}$	w	46 273	{46 230	22 $\bar{5}$
vw	27 114	25 937	022			{46 295	42 $\bar{5}$
vw	27 285	27 116	514	m	46 717	46 669	11 $\bar{6}$
w	27 538	27 311	601	vw	47 131	47 104	314
w		27 554	604	vw	47 374	47 352	80 $\bar{5}$

As pointed out by Furusetth and Kjekshus<sup>3</sup> there are slight mistakes in the interatomic distances listed by Boller and Parthé.<sup>5</sup> With the present lattice dimensions the shortest interatomic distances are (in Å):

Table 7. Guinier photograph data of TaSb<sub>3</sub> taken with strictly monochromatized CuK $\alpha_1$ -radiation.

$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$	$I_{\text{obs}}$	$\sin^2\theta \times 10^5$		$hkl$
	obs	calc			obs	calc	
w	1 160	1 160	001	w	25 641	25 643	223
m	2 312	2 309	201	w	26 054	26 051	315
vw	3 056	3 053	200	w	27 099	27 098	422
vw	3 885	3 884	202	w	27 417	{27 426	421
w	5 228	5 229	110	vw		{27 473	600
vw	5 434	5 437	111	w	27 596	27 594	114
vw	6 132	6 116	201	w	27 905	27 911	605
vst	7 341	7 341	111	m	28 301	28 302	023
m	7 774	7 779	203	w	29 087	29 089	423
st	7 965	7 965	112	w	29 464	{29 464	511
w	9 228	9 234	402	w		{29 465	115
w	9 567	9 562	401	m	30 075	30 074	420
m	9 631	9 638	311	vw	30 335	30 340	313
m	10 253	10 262	312	vw	31 112	31 118	406
m	10 436	10 438	003	vw	31 850	31 858	224
m	11 222	11 225	403	m	32 310	32 308	713
st	11 330	11 334	310	w	33 375	{33 383	206
m	11 771	11 772	112	w		{33 400	424
m	12 204	12 210	400	vw	34 073	34 072	403
w	12 808	12 812	113	vw	35 035	35 042	421
m	13 201	13 205	313	vw	36 432	{36 421	024
w	13 997	13 994	204	w		{36 433	803
vw	15 339	15 350	311	m	36 743	36 740	516
vw	17 171	17 178	401	m	37 072	37 067	223
m	17 857	17 864	020	vw	37 722	37 703	512
w	18 446	18 468	314	vw	38 239	38 249	802
w	18 663	18 664	512	w	38 991	38 985	115
w	19 203	19 203	203	vw	39 771	39 759	805
vw	19 704	19 704	513	vw	40 019	40 031	425
vw	19 972	{19 945	511	vw	40 408	40 393	225
vw		{19 979	114	vw	40 807	40 785	621
vw	20 164	20 172	221	vw	41 051	41 048	624
vw	20 681	20 689	602	w	41 275	{41 272	116
w	21 676	21 685	312	w		{41 315	314
vw	22 167	22 167	405	w	41 864	41 860	710
vw	22 534	{22 503	022	w		{42 329	422
vw		{22 529	225	vw	42 364	{42 385	801
vw	23 187	23 184	604	w		{42 388	407
vw	23 548	23 544	510	w	43 076	43 069	131
vw	24 002	23 978	514	w		{43 632	716
vw	24 443	24 465	402	w	43 655	{43 692	132

Ta(As) — 4 Ta(As): 3.379  
 — 4 Ta(As): 3.435  
 — 2 As(Ta): 2.591  
 — 4 As(Ta): 2.615

Table 8. Lattice constants of TaAs<sub>2</sub> and TaSb<sub>2</sub>.

Phase	<i>a</i> (Å)	<i>b</i> (Å)	<i>c</i> (Å)	β (°)
TaAs <sub>2</sub> <sup>a</sup>	9.3385	3.3851	7.7568	119.70
TaAs <sub>2</sub> <sup>b</sup>	9.350 ± 0.002	3.388 ± 0.002	7.760 ± 0.002	119.75 ± 0.08
TaSb <sub>2</sub> <sup>a</sup>	10.2218	3.6447	8.2915	120.39

<sup>a</sup> This study. <sup>b</sup> Quoted from Saini *et al.*<sup>6</sup>

Four phases are known with the NbAs-type structure, *i.e.* NbAs,<sup>3,5,6</sup> NbP,<sup>5,19</sup> TaP,<sup>5,19</sup> and TaAs.<sup>5,6</sup>

*The TaAs<sub>2</sub> and TaSb<sub>2</sub> phases.* The phases TaAs<sub>2</sub> and TaSb<sub>2</sub> were easily achieved by heating mixtures of the elements. Guinier photographs (Tables 6 and 7) of the stoichiometric TaAs<sub>2</sub> and TaSb<sub>2</sub> samples were indexed monocrinally. The unit cell dimensions will be seen from Table 8, where also the values previously determined for TaAs<sub>2</sub> by Saini *et al.*<sup>6</sup> are listed. Although the present values do not fall within the error limits given by them, the accordance is considered satisfactory.

On the basis of the observed densities 10.26 g cm<sup>-3</sup> (TaAs<sub>2</sub>) and 10.53 g cm<sup>-3</sup> (TaSb<sub>2</sub>) the unit cells contain 4 TaX<sub>2</sub>-groups (*Z*<sub>c</sub> = 3.97 for TaAs<sub>2</sub> and *Z*<sub>c</sub> = 3.98 for TaSb<sub>2</sub>). The calculated densities from the X-ray data are 10.33 g cm<sup>-3</sup> (TaAs<sub>2</sub>) and 10.58 g cm<sup>-3</sup> (TaSb<sub>2</sub>).

Guinier photographs of TaAs<sub>2</sub> and TaSb<sub>2</sub> in samples where these phases are in equilibrium with phases of higher and lower metal content, give no significant variation of the lattice constants. The ranges of homogeneity are thus rather narrow.

The systematic extinctions are of the type

$$hkl \text{ absent when } h + k = 2n + 1$$

Characteristic space groups are accordingly *C2*, *Cm*, and *C2/m*.

Obvious relationships in lattice dimensions and possible space groups indicated that the structure of both phases might be of the NbAs<sub>2</sub>-type.<sup>1,4</sup> To confirm this suggestion, a set of calculated structure factors (based on the parameter values (rounded off) from NbAs<sub>2</sub> and NbSb<sub>2</sub>) were compared with those observed. All possible reflections in the Guinier photographs were included in the calculation. The obtained agreements were surprisingly good and the proposed NbAs<sub>2</sub>-type structure must be considered proved.

In terms of space group *C2* all atoms are in

$$(c) \ x, y, z; \bar{x}, y, \bar{z}; \frac{1}{2} + x, \frac{1}{2} + y, \frac{1}{2} + z; \frac{1}{2} - x, \frac{1}{2} + y, \frac{1}{2} - z$$

with:

$$\begin{array}{l} \text{TaAs}_2: \quad x_{\text{Ta}} = 0.34, \quad y_{\text{Ta}} = \frac{1}{2}, \quad z_{\text{Ta}} = 0.30 \\ \quad \quad x_{\text{AsI}} = 0.09, \quad y_{\text{AsI}} = 0.49, \quad z_{\text{AsI}} = 0.39 \\ \quad \quad x_{\text{AsII}} = 0.14, \quad y_{\text{AsII}} = 0.07, \quad z_{\text{AsII}} = 0.03 \end{array}$$



$$\begin{aligned} \text{TaSb}_2: \quad x_{\text{Ta}} &= 0.34, \quad y_{\text{Ta}} = \frac{1}{2}, \quad z_{\text{Ta}} = 0.30 \\ x_{\text{SbI}} &= 0.10, \quad y_{\text{SbI}} = 0.5, \quad z_{\text{SbI}} = 0.39 \\ x_{\text{SbII}} &= 0.14, \quad y_{\text{SbII}} = 0.0, \quad z_{\text{SbII}} = 0.03 \end{aligned}$$

The parameter values are considered to be quite good.

The interatomic distances are listed in Table 9. For a detailed description of the  $\text{NbAs}_2$ -type structure reference is made to Furuseth and Kjekshus.<sup>4</sup> Six phases are known at present with the  $\text{NbAs}_2$ -type structure, *i.e.*  $\text{NbAs}_2$ ,<sup>1,4</sup>  $\text{NbSb}_2$ ,<sup>1,4</sup>  $\text{TaAs}_2$ ,  $\text{TaSb}_2$ ,  $\text{MoAs}_2$ ,<sup>20</sup> and  $\text{WAs}_2$ .<sup>21</sup>

Table 9. Interatomic distances in  $\text{TaAs}_2$  and  $\text{TaSb}_2$  (Å).

$\text{TaAs}_2$ : Ta	— 3 Ta	: 3.06; 3.385 (2)
	— 5 As <sub>I</sub>	: 2.66; 2.70; 2.72; 2.75; 2.76
	— 3 As <sub>II</sub>	: 2.47; 2.67; 2.78
As <sub>I</sub>	— 5 Ta	: 2.66; 2.70; 2.72; 2.75; 2.76
	— 5 As <sub>I</sub>	: 2.93; 3.11 (2); 3.385 (2)
	— 6 As <sub>II</sub>	: 3.21; 3.37; 3.49; 3.63; 3.73; 3.93
As <sub>II</sub>	— 3 Ta	: 2.47; 2.67; 2.78
	— 6 As <sub>I</sub>	: 3.21; 3.37; 3.49; 3.63; 3.73; 3.93
	— 5 As <sub>II</sub>	: 2.42; 2.87 (2); 3.385 (2)
$\text{TaSb}_2$ : Ta	— 3 Ta	: 3.28; 3.645 (2)
	— 5 Sb <sub>I</sub>	: 2.90; 2.95 (2); 2.99 (2)
	— 3 Sb <sub>II</sub>	: 2.81 (2); 2.85
Sb <sub>I</sub>	— 5 Ta	: 2.90; 2.95 (2); 2.99 (2)
	— 5 Sb <sub>I</sub>	: 3.22 (2); 3.36; 3.645 (2)
	— 6 Sb <sub>II</sub>	: 3.58 (2); 3.69 (2); 4.10; 4.15
Sb <sub>II</sub>	— 3 Ta	: 2.81 (2); 2.85
	— 6 Sb <sub>I</sub>	: 3.58 (2); 3.69 (2); 4.10; 4.15
	— 5 Sb <sub>II</sub>	: 2.65; 3.12 (2); 3.645 (2)

*Magnetic properties.* The magnetic susceptibilities of  $\text{Ta}_5\text{Sb}_4$ ,  $\text{TaAs}$ ,  $\text{TaAs}_2$ , and  $\text{TaSb}_2$  were measured at temperatures between 90 and 725°K. (The  $\text{Ta}_3\text{Sb}$

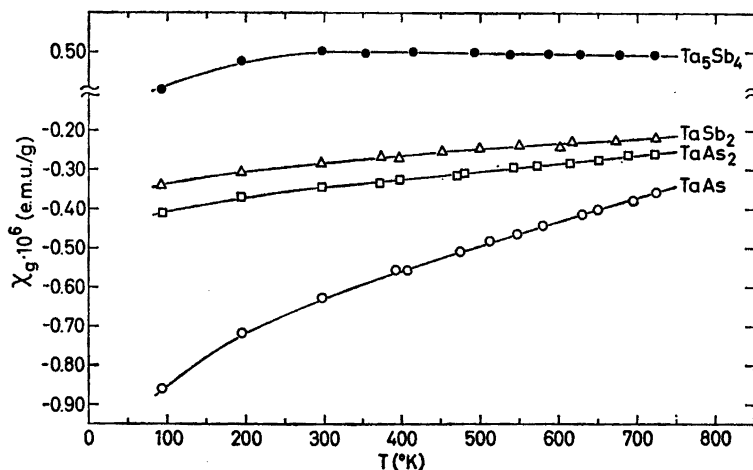


Fig. 1. The magnetic susceptibilities of  $\text{TaAs}$ ,  $\text{TaAs}_2$ ,  $\text{Ta}_5\text{Sb}_4$ , and  $\text{TaSb}_2$  as a function of temperature.

phase was considered too impure to be included among the results of magnetic susceptibility measurements in this communication.) The results, *cf.* Fig. 1, show either diamagnetic or weak almost temperature independent susceptibilities.

The expected diamagnetism resulting from the atomic cores in the three diamagnetic substances TaAs, TaAs<sub>2</sub>, and TaSb<sub>2</sub> is of the same size as observed. The core contributions are  $-0.31 \times 10^{-6}$ ,  $-0.43 \times 10^{-6}$ , and  $-0.48 \times 10^{-6}$  e.m.u. per gram TaAs, TaAs<sub>2</sub>, and TaSb<sub>2</sub>, respectively. (These values were calculated from the diamagnetic corrections  $-14 \times 10^{-6}$  e.m.u. per mole Ta<sup>5+</sup> according to Klemm<sup>22</sup> and  $-64.6 \times 10^{-6}$  e.m.u. per mole As<sup>3-</sup> and  $-94.4 \times 10^{-6}$  e.m.u. per mole Sb<sup>3-</sup> according to Angus.<sup>23</sup>) By subtracting these values from the susceptibilities measured the total susceptibilities of localized, non-bonding electrons, valence electrons and conduction electrons are obtained. The measurements show that no unpaired *d*-electrons are present on the tantalum atoms, and suggest a covalent type of bonding in TaAs, TaAs<sub>2</sub>, and TaSb<sub>2</sub>.

The core contribution is  $-0.32 \times 10^{-6}$  e.m.u. per gram Ta<sub>5</sub>Sb<sub>4</sub>. Ta<sub>5</sub>Sb<sub>4</sub> is expected to have a metallic type of bonding and the observed paramagnetism should consequently be associated with the paramagnetism of the spin of the conduction electrons.

When it is added that magnetic measurements carried out on the impure Ta<sub>3</sub>Sb sample indicate weak temperature independent paramagnetism, the almost complete identity in magnetic properties between corresponding niobium and tantalum arsenide and antimonide phases becomes evident.

#### DISCUSSION

A discussion of the bonding in the metallic like phases Ta<sub>3</sub>Sb and Ta<sub>5</sub>Sb<sub>4</sub> must await further experimental data.

The bonding in TaAs, TaAs<sub>2</sub>, and TaSb<sub>2</sub> is conveniently discussed in terms of the general (8-N) rule:<sup>24-27</sup>

$$(n_v + b_a - b_c)/n_a = 8$$

where per formula unit  $n_v$  is the number of valence electrons,  $n_a$  the number of anions,  $b_a$  the number of valence electrons involved in anion-anion bond formation, and  $b_c$  is the number of valence electrons involved in cation-cation bond formation (together with any unshared electrons on the cations). (This rule is a mathematical formulation of the need for all the anions in a crystal to possess a complete octet.<sup>27</sup>)

In the crystal structure of TaAs (similar for the isostructural phases NbAs, NbP, and TaP) there are no short As-As or Ta-Ta distances which can be associated with anion-anion or cation-cation bonds. According to its structure TaAs is therefore a normal valence compound. Furthermore, TaAs is diamagnetic (the same is found for NbAs<sup>2</sup>) and each tantalum atom must consequently contribute 5, 3, or 1 electron to  $n_v$ . Each arsenic atom provides 5 electrons. In order to comply with the rule three of the outer electrons from the tantalum atoms must therefore participate in the chemical bonding leaving an inert electron pair on each tantalum atom. Data for the electrical

Table 10. Application of the general (8-N) rule on some  $MX_2$ -phases.

Phase	Structure type	$n_c/n_a$	$b_a/n_a$	$b_c/n_a$	Anion substructure	Reference
$TlSb_2$ , $HfAs_2$ $ZrAs_2$ , $TbSb_2$	$CuAl_2$ $PbCl_2$ $Cu_3Sb$	7 7 7	1 $\frac{1}{2}$	$-\frac{1}{2}$ -1	$X-X$ pairs $X-X$ pairs between I-I	Pearson <sup>28</sup> Trzebiatowski <i>et al.</i> <sup>29</sup> Jeltschko and Nowotny <sup>30</sup> Pearson <sup>28</sup>
$VSb_2$ , $NbAs_2$ , $NbSb_2$ , $TaAs_2$ , $TaSb_2$	$CuAl_2$ $NbAs_2$	$7\frac{1}{2}$ $7\frac{1}{2}$	1 $\frac{1}{2}$	$\frac{1}{2}$	$X-X$ pairs $X-X$ pairs between II-II	Pearson <sup>28</sup> Furuseth and Kjekshus <sup>4</sup>
$CrSb_2$ , $MoP_2$ , $WP_2$ $MoAs_2$ , $WAs_2$ $UP_2$ , $UAs_2$ , $USb_2$ , $UBi_2$	Marcasite $MoP_2$ $NbAs_2$ $Cu_3Sb$	8 8 8 8	1 1 $\frac{1}{2}$	1 1 $\frac{1}{2}$	$X-X$ pairs $X-X$ pairs between I-II $X-X$ pairs between II-II	Pearson <sup>28</sup> Rundqvist and Lundström <sup>31</sup> Jensen and Kjekshus <sup>20</sup> Skansen and Kjekshus <sup>21</sup> Pearson <sup>28</sup>

conductivities or optical properties of these substances are needed before this suggestion can be fully accepted.

The general (8-N) rule has previously been applied to pnictogen phases with composition  $MX_2$ ,  $M$  being a transition metal from subgroup IV, V, or VI.<sup>4</sup> Table 10 (which is quoted from Furusetth and Kjekshus<sup>4</sup> with some additions) shows the partition of  $n_e/n_a$  (based on valences corresponding to the group numbers)  $b_a/n_a$  (according to the crystal structures) and  $b_c/n_a$  (assuming the general (8-N) rule satisfied) for the considered compounds.

The new phases included in Table 10 are  $TaAs_2$ ,  $TaSb_2$ ,  $MoAs_2$ , and  $WAs_2$ , with the  $NbAs_2$ -type structure. The rule appears to be fulfilled as far as  $TaAs_2$  and  $TaSb_2$  are concerned, whose  $b_c/n_a = 0$  is consistent with the diamagnetism observed. For  $MoAs_2$  and  $WAs_2$ , however, the calculated  $b_c/n_a = \frac{1}{2}$  suggests that one electron should be localized on each metal atom. This is in disagreement with the observed diamagnetism for  $MoAs_2$ <sup>20</sup> and  $WAs_2$ .<sup>21</sup> We are awaiting the gathering of more data on these substances before any conclusions can be drawn.

*Acknowledgement.* The authors wish to thank Professor Haakon Haraldsen for his kind interest in this study and for placing laboratory facilities at their disposal.

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Received October 2, 1964.

*Acta Chem. Scand.* **19** (1965) No. 1